

38th International Conference on Plasma Science (ICOPS)  
and 24th Symposium on Fusion Engineering (SOFE)  
Chicago, USA June 26 - 30, 2011.

# Li & Pb MD Simulations

Alberto Fraile

Dr. Santiago Cuesta

Prof. J. Manuel. Perlado

Instituto Fusion Nuclear Madrid Spain

Dr. Alfredo Caro

Los Alamos National Laboratory, NM, USA



POLITÉCNICA



# Outline

- Motivation
- MD, potentials and methods
- Results
- Conclusions
- LiPb EAM/alloy/cd results
- Future work

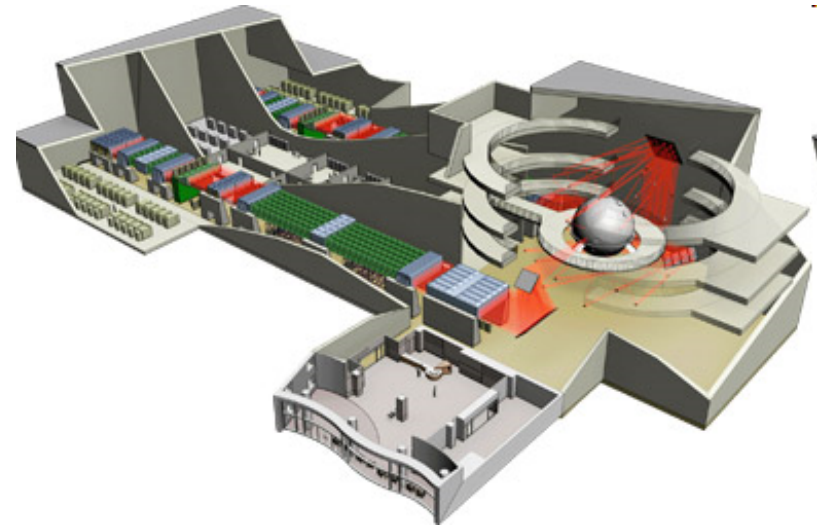
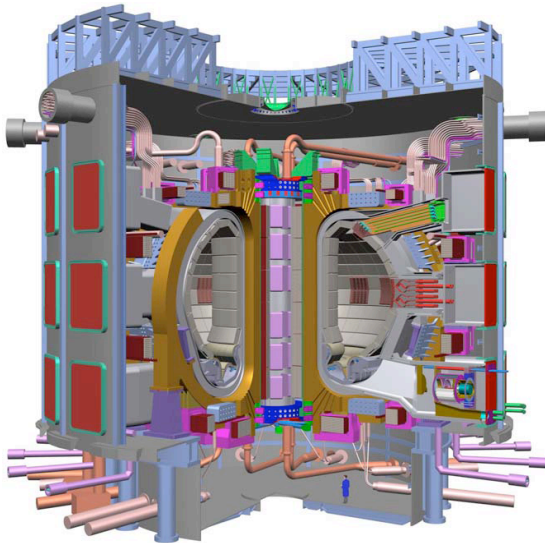


# ■ **Motivation**

- MD, potentials and methods
- Results
- Conclusions
- LiPb EAM/alloy/cd results
- Future work

# Fusion Reactor Technologies

- ITER, HiPER etc



We will need a blanket...

# Blanket solutions I

## ■ LiPb & other possible candidates

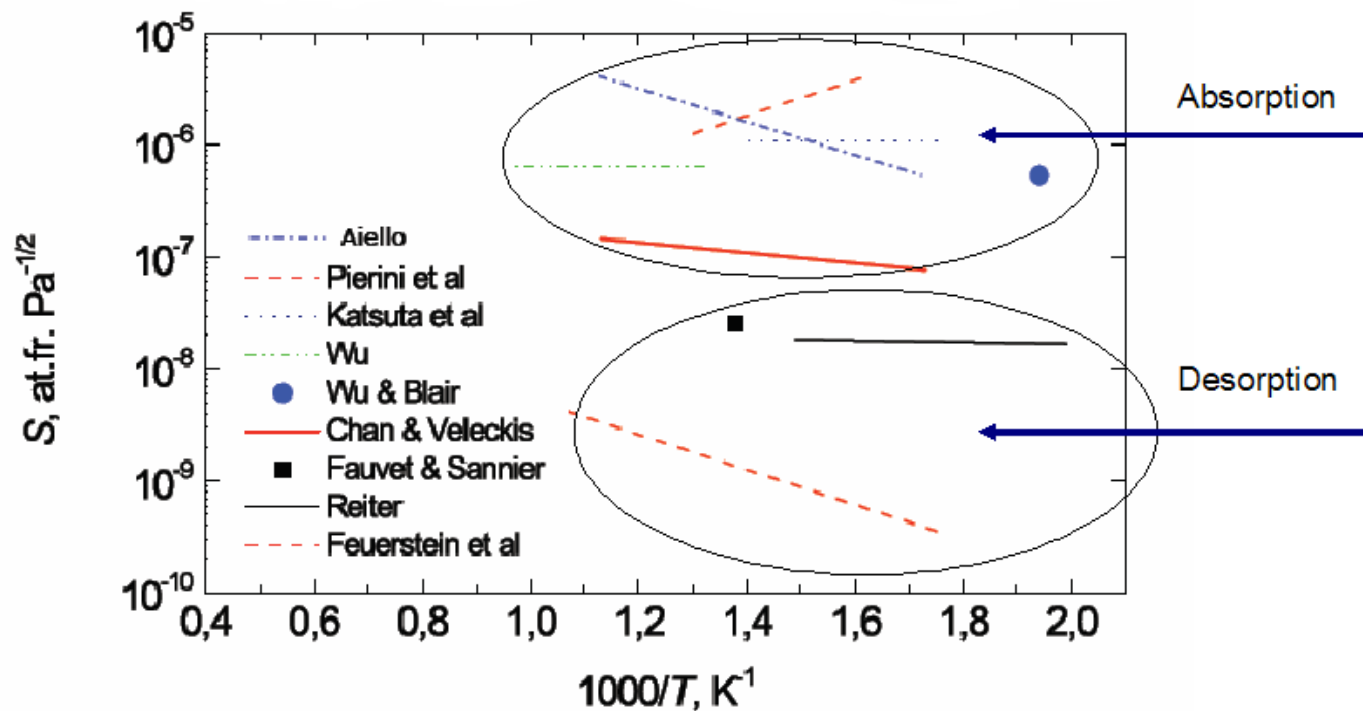
<b>Liquid breeder</b>	<b>Li</b>	<b>Li<sub>17</sub>Pb<sub>83</sub></b>	<b>Flibe</b>	<b>Li<sub>20</sub>Pb<sub>80</sub></b>
Melting point (°C)	180	235	459	320
Density (g/cm <sup>3</sup> ) 873K	0.48	8.98	2.0	6.0
Li Density (g/cm <sup>3</sup> ) 873K	0.48	0.061	0.28	0.09
Breeding property	Good	Fairly good	Neutron multiplier required	Neutron multiplier required
Chemical stability	Active	Middle	Almost stable	Almost stable
Corrosion	Severe	Middle	HF exist severe	?
Tritium release form	HT, T <sub>2</sub>	HT, T <sub>2</sub>	HT, T <sub>2</sub> TF	HT, T <sub>2</sub>

# Blanket solutions II

## ■ LiPb & other possible candidates

Liquid breeder	Li	Li <sub>17</sub> Pb <sub>83</sub>	Flibe	Li <sub>20</sub> Pb <sub>80</sub>
Tritium solubility (atom frac Pa <sup>-0.5</sup> T=873K)	Very high 7.49x10 <sup>-3</sup>	Very low 1.93x10 <sup>-8</sup>	Very low HT/T <sub>2</sub> 1.77x10 <sup>-11</sup> Pa <sup>1</sup> TF 1.77x10 <sup>-11</sup> Pa <sup>-1</sup>	Middle 2x10 <sup>-7</sup> -1x10 <sup>-5</sup>
Tritium diffusivity order (m <sup>2</sup> /s) (873K)	Relatively high 10 <sup>-9</sup>	Relatively high 10 <sup>-9</sup>	Relatively high 10 <sup>-9</sup>	Relatively high 10 <sup>-9</sup>
Thermal conductivity	Li>Li <sub>20</sub> Sn <sub>80</sub> > Li <sub>17</sub> Pb <sub>83</sub> > Flibe			
Dynamic viscosity	Flibe>Li <sub>20</sub> Sn <sub>80</sub> ~Li <sub>17</sub> Pb <sub>83</sub> >Li			

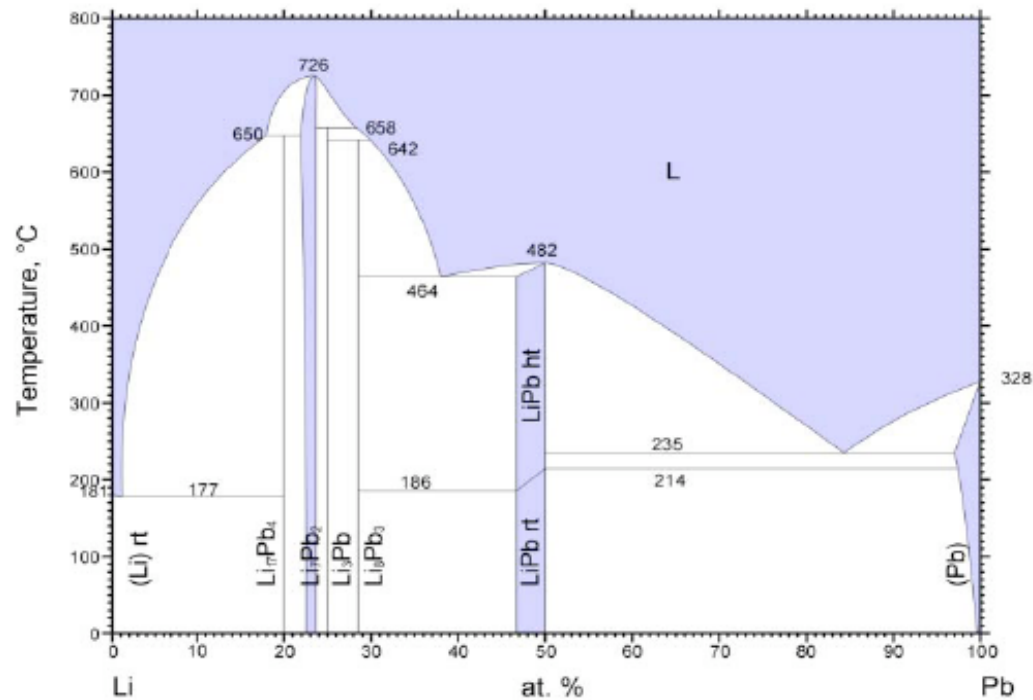
# Tritium (& He) behaviour



**Figure 1:** Solubility database is inadequate for design. Scatter reflects experimental approaches and measurement techniques applied. Knowledge of dynamic transport properties (diffusion, mass transfer, interface processes) is much more limited [1].

# LiPb system

## ■ Phase diagram





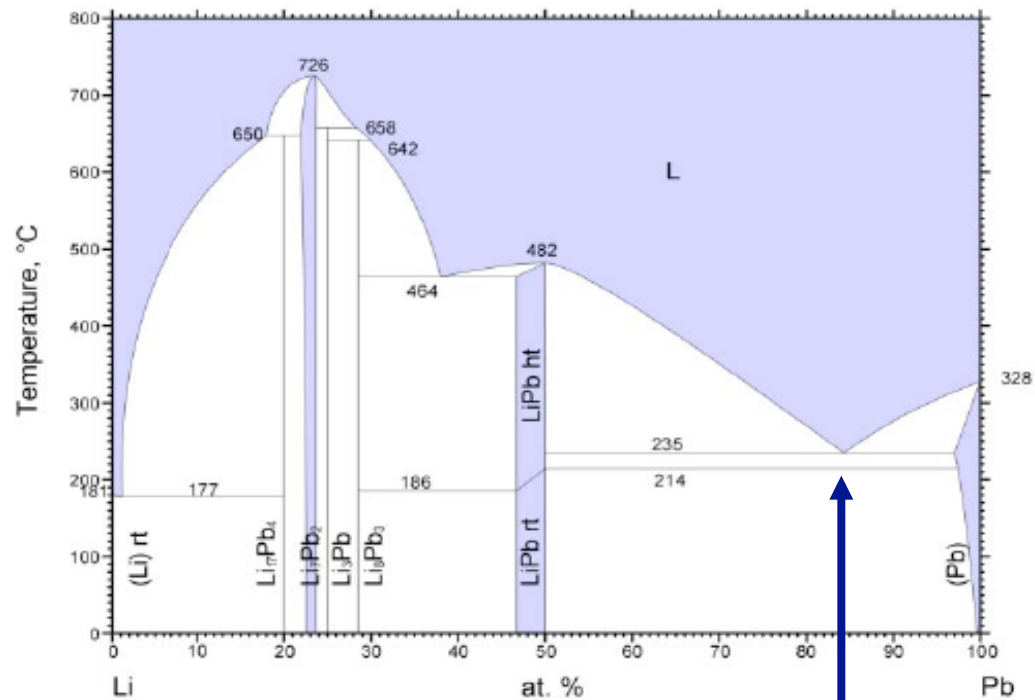
# LiPb system

## ■ Phase diagram

Eutectic point is still to be determined.

Structural properties of liquid phase remain unclear.

Magnetic field effects?



Eutectic Li = 15,7 – 17 % (to be determined!)



- Motivation
- **MD, potentials and methods**
- Results
- Conclusions
- LiPb EAM/alloy/cd results
- Future work

# Li & Pb EAM potentials

The total internal energy of the system of  $N$  atoms is described as the energy required to embed these  $N$  atoms into the homogeneous electron gas caused by surrounding atoms plus a correction of energy from two-body interactions. Thus this total energy can be expressed as:

$$E_i = \frac{1}{2} \sum_{i,j,i \neq j} \phi_{ij}(r_{ij}) + \sum_i F_i(\rho_i)$$

where  $\phi_{ij}$  represents the pair energy between atoms  $i$  and  $j$  separated by  $r_{ij}$ , and  $F_i$  stands for the embedding energy to embed an atom  $i$  into a local site with electron density  $\rho_i$ .

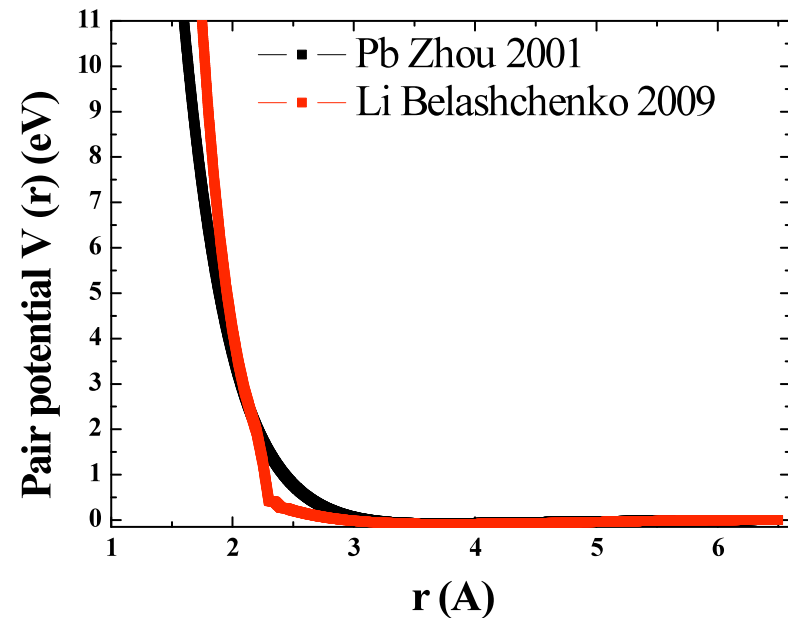


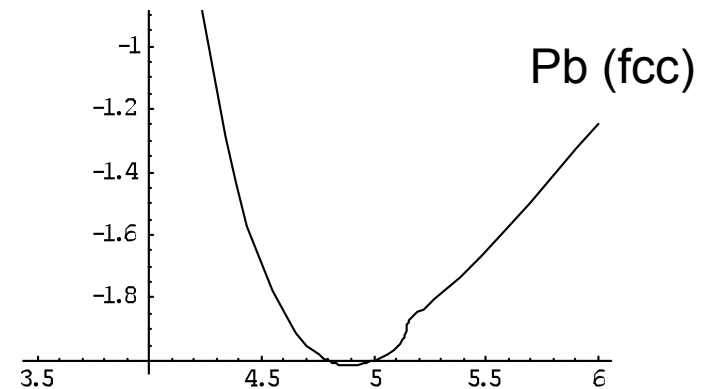
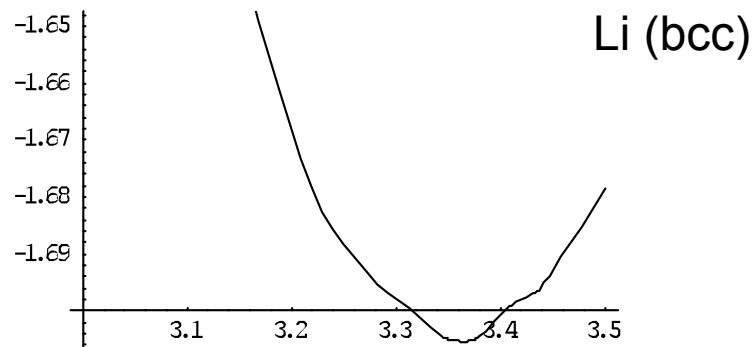
Fig. 2. Pair potentials for lead and lithium following references [3], [4].



- Motivation
- MD, potentials and methods
- **Results**
- Conclusions
- LiPb EAM/alloy/cd results
- Future work

# Testing Li & Pb potentials I

## ■ Cohesive energy & Lattice parameter

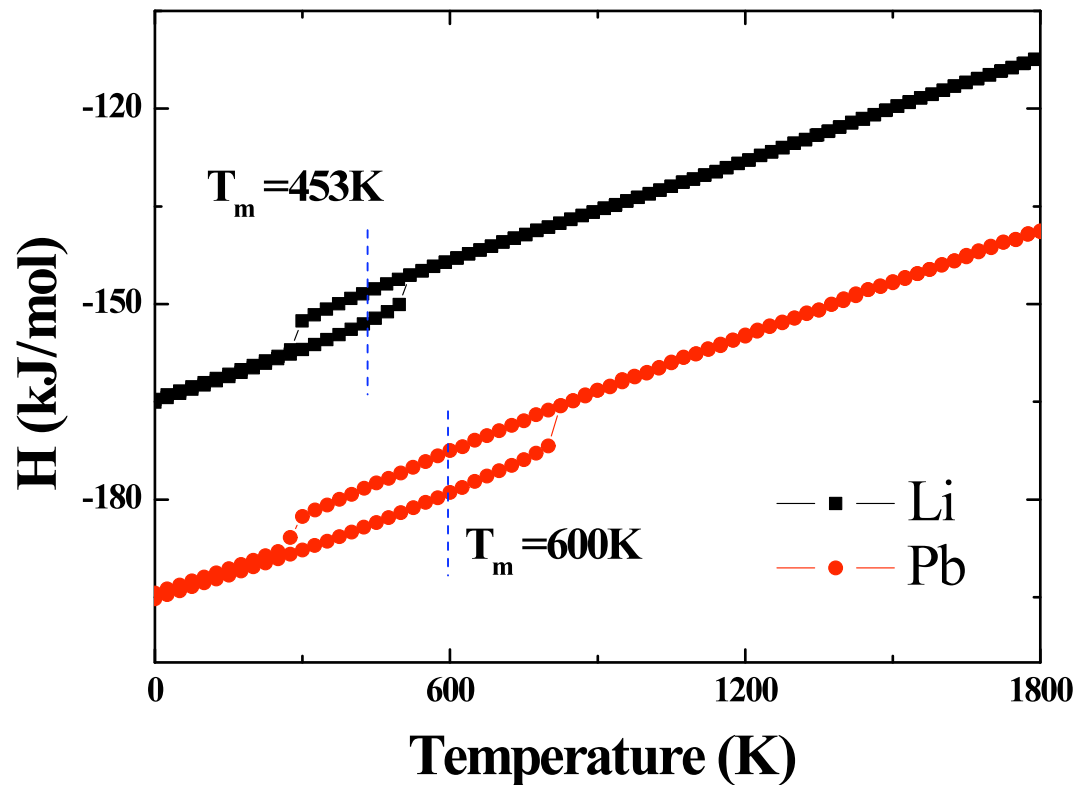


Element	Theoretical calculation		Experiment	
	$E_c$ (eV/at)	Cell parameter (A)	$E_c$ (eV/at)	Cell parameters (A)
Pb (fcc)	-2.014	a= 4.894	-2.03	a= 4.9095
Li (bcc)	-1.705	a= 3.3648	-1.63	a= 3.355

Table 1. Calculated cohesive energy,  $E_c$ , and lattice parameter,  $a$ , for Pb and Li in fcc and bcc phases. All values have been calculated **using Pb and Li EAM potentials** in the **effective representation**. The results are exactly the same that those obtained with the original potentials.

# Testing Li & Pb potentials II

- Melting point



Experimental:  $T_m(\text{Pb}) = 600 \text{ K}$ ,  $T_m(\text{Li}) = 453 \text{ K}$

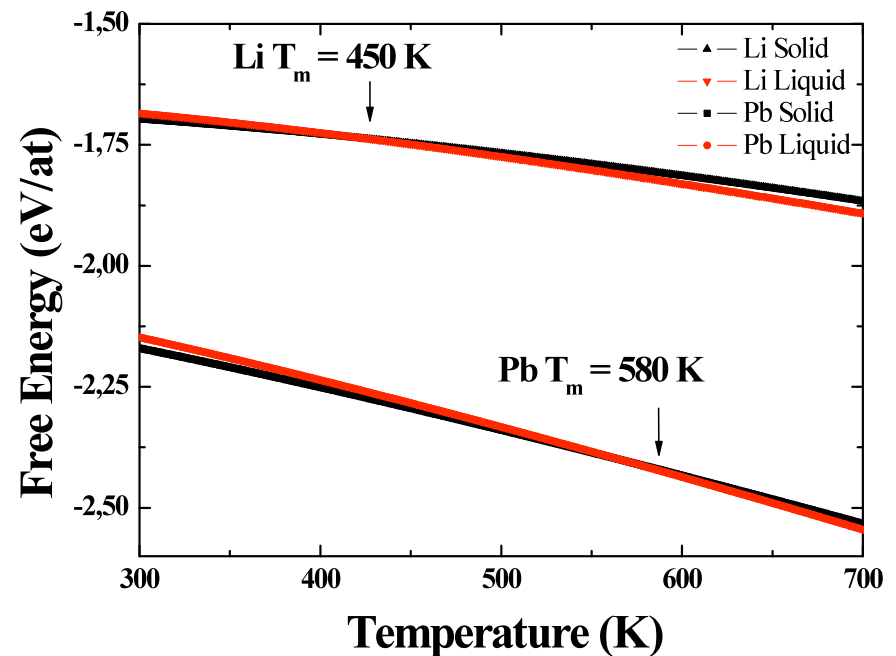
# Testing Li & Pb potentials II

## ■ Melting point

Free energies can be calculated using the Gibbs-Duhem integral:

$$f(T) = f(T_0) \frac{T}{T_0} - T \int_{T_0}^T \frac{h(\tau)}{\tau^2} d\tau$$

where  $h(\tau)$  is the enthalpy per particle. The coupling-constant integration method, or switching Hamiltonian method, [5] is used to calculate  $f(T_0)$  [6].



Melting point from free energy calculations.  
Experimental:  $T_m(\text{Pb}) = 600 \text{ K}$ ,  $T_m(\text{Li}) = 453 \text{ K}$

# Testing Li & Pb potentials III

- Structural properties

$$R(r) = \frac{1}{N} \sum_{\nu} \sum_{\mu} \frac{b_{\nu} b_{\mu}}{\langle b \rangle^2} \delta(r - r_{\nu\mu})$$

$$\rho(r) = \frac{1}{4\pi r^2} R(r)$$

Radial distribution function  $g(r) = \rho(r)/\rho_0 = \frac{R(r)}{4\pi\rho_0 r^2}$     thus  $g(r) \rightarrow 1$  for  $r \rightarrow \infty$

$$G(r) = 4\pi r \rho_0 (g(r) - 1)$$

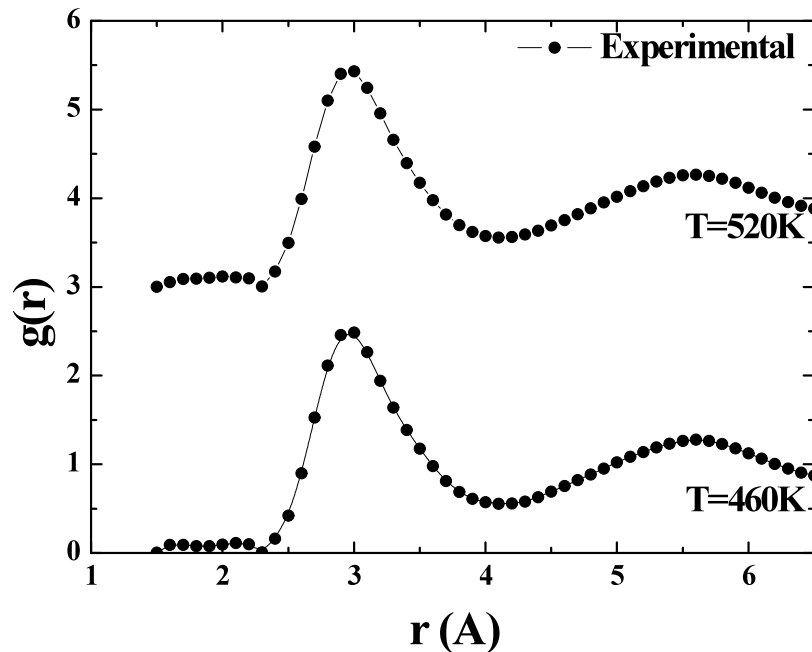
$$G(r) = \frac{2}{\pi} \int_0^{\infty} Q [S(Q) - 1] \sin(Qr) dQ$$

Total structural factor  $S(Q) = 1 + \frac{1}{Q} \int_0^{\infty} G(r) \sin(Qr) dr$

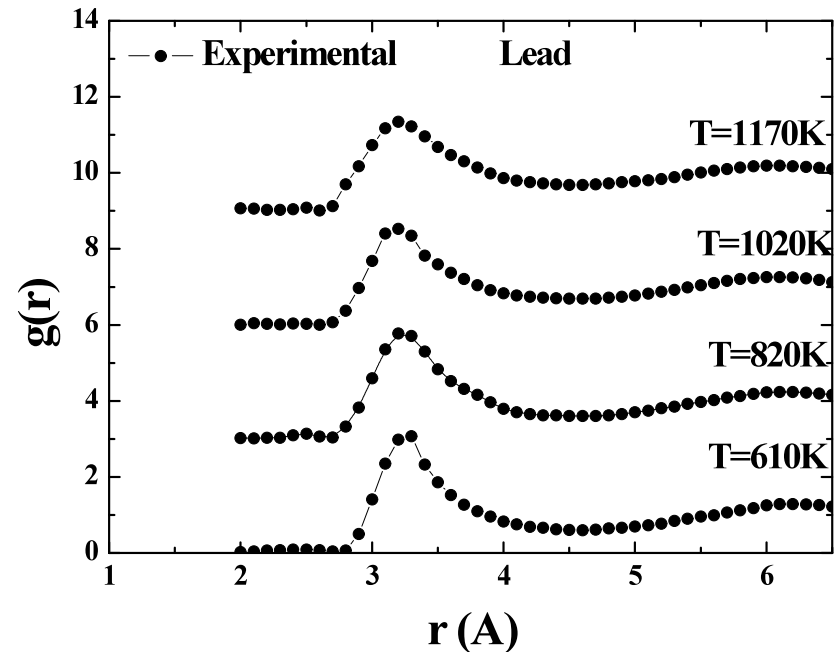


# Testing Li & Pb potentials IIb

## ■ Structural properties ( $g(r)$ )



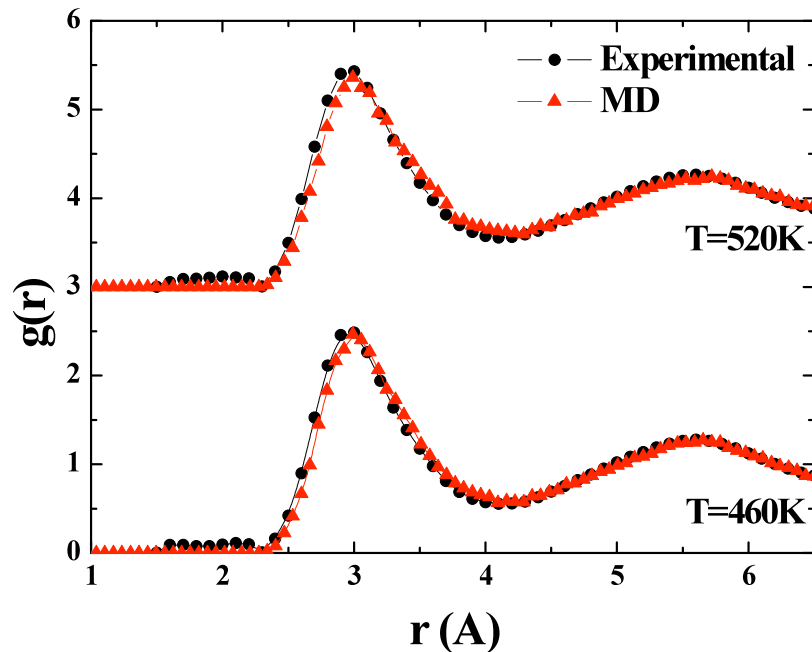
Lithium  $g(r)$  experimental results.



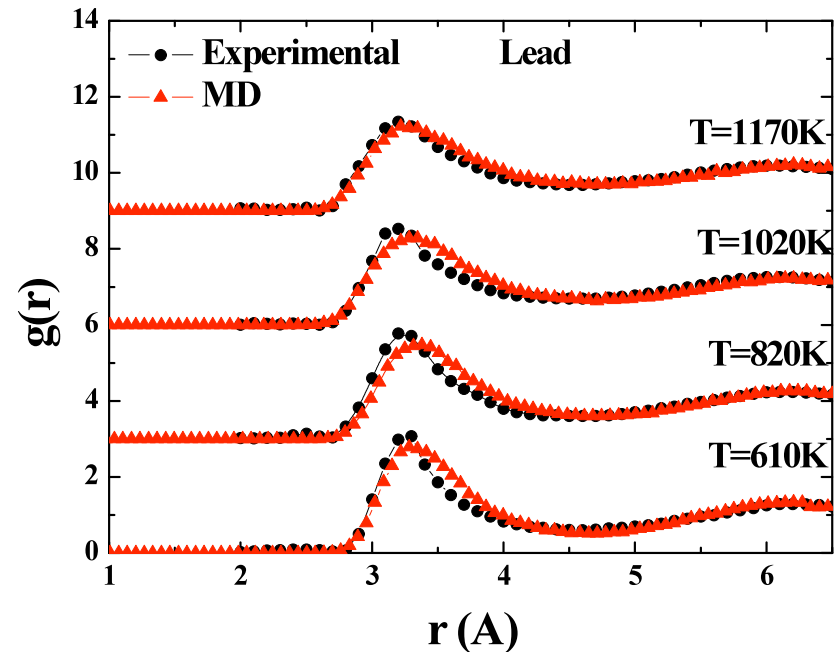
Lead  $g(r)$  experimental data.

# Testing Li & Pb potentials IIb

## ■ Structural properties ( $g(r)$ )



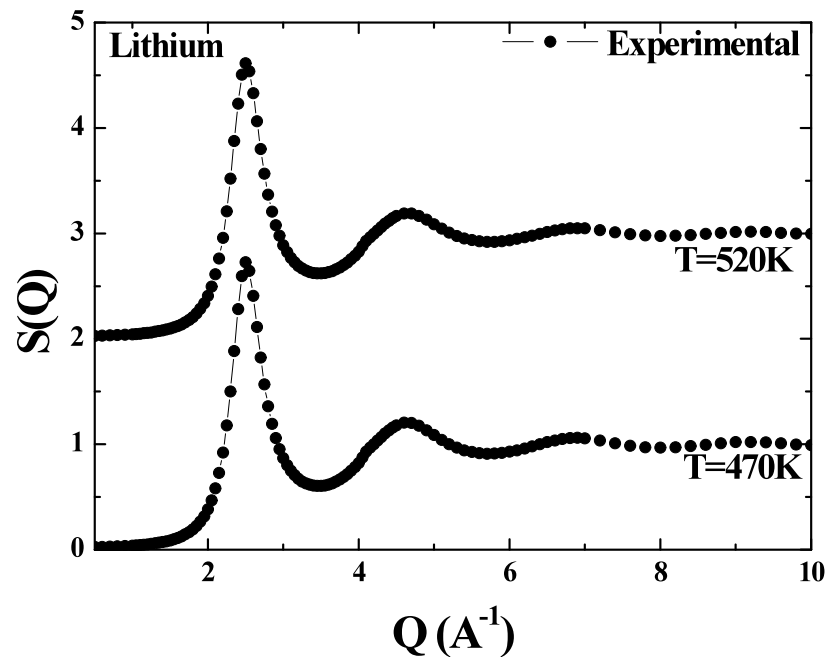
Lithium  $g(r)$  calculated (red triangles) compared with experimental results (black circles).



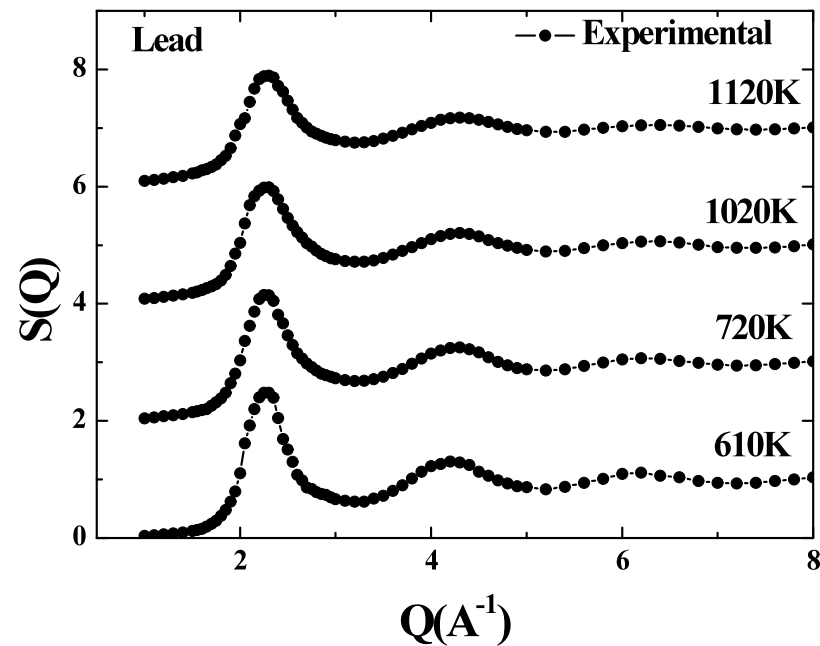
Lead  $g(r)$  calculated (red triangles) compared with experimental (black circles)

# Testing Li & Pb potentials IIc

## ■ Structural properties ( $S(Q)$ )



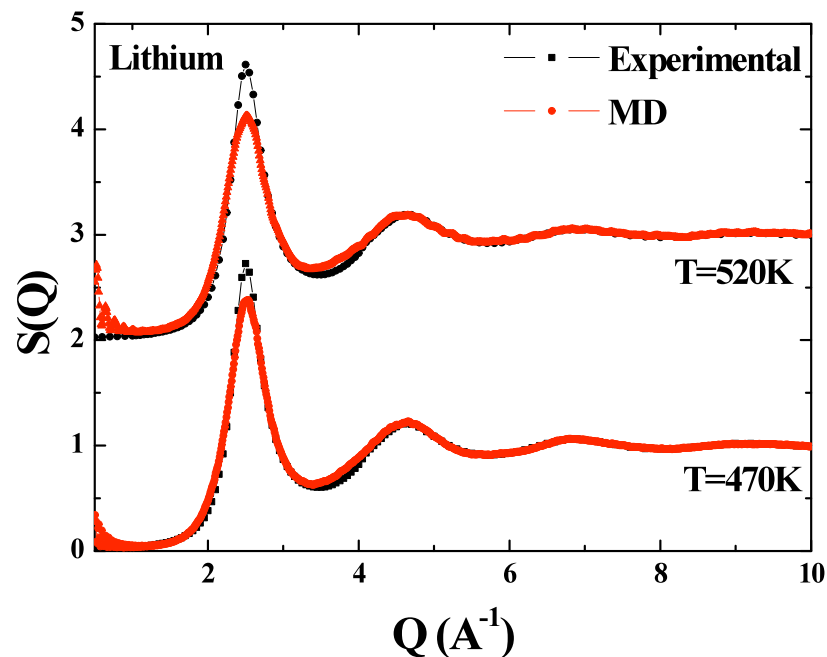
Lithium  $S(Q)$  experimental data.



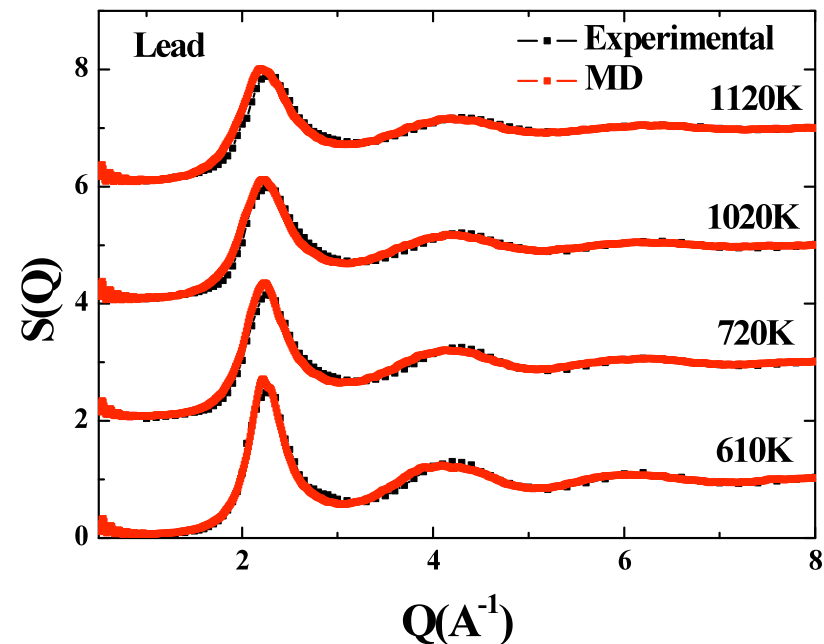
Lead  $S(Q)$  experimental data.

# Testing Li & Pb potentials IIc

## ■ Structural properties ( $S(Q)$ )



Lithium  $S(Q)$  calculated (red) compared with experimental results (black).



Lead  $S(Q)$  calculated (red) compared with experimental (black)

# Testing Li & Pb potentials IV

## ■ Density ( $\rho$ )

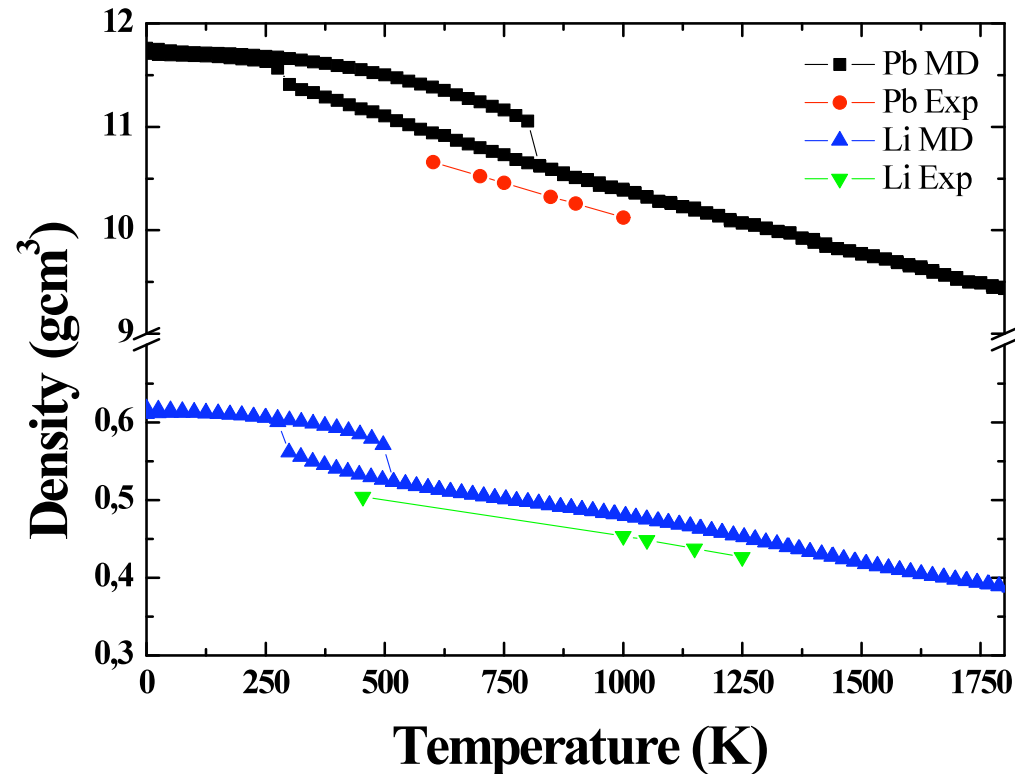


Figure 3. Density,  $\rho$ , calculated from MD simulations (black squares for lead and blue triangles for lithium) compared with experimental values in liquid phase, red circles for Pb and inverted green triangles for Li.

# Testing Li & Pb potentials IV

## Density ( $\rho$ ) & $d\rho/dT$

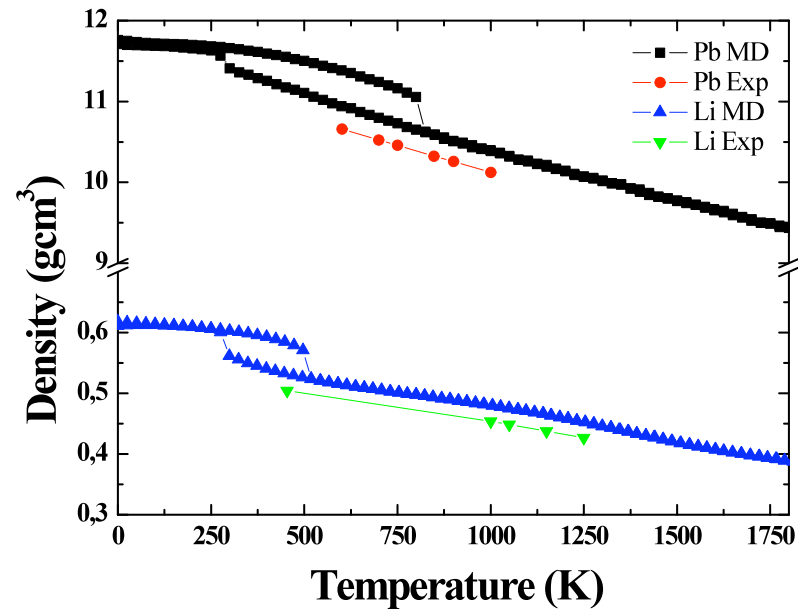


Figure 3. Density,  $\rho$ , calculated from MD simulations (black squares for lead and blue triangles for lithium) compared with experimental values in liquid phase, red circles for Pb and inverted green triangles for Li.

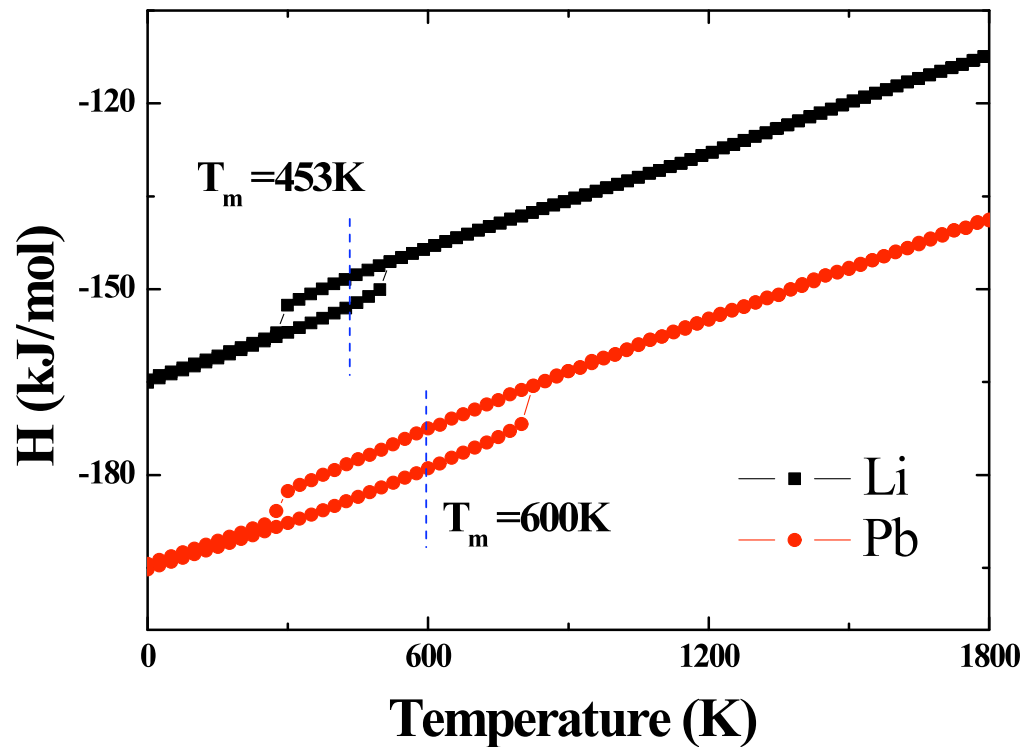
	Experimental		MD	
	$\rho$ (Liq Temp) g/cm <sup>3</sup>	$\partial\rho/dT$ (g/ cm <sup>3</sup> K)	$\rho$ (Liq Temp) g/cm <sup>3</sup>	$\partial\rho/dT$ (g/ cm <sup>3</sup> K)
<b>Li</b>	0,51	$-9,6 \times 10^{-5}$	0,53	$-8,6 \times 10^{-5}$
<b>Pb</b>	10,64	-0.0013	10,96	-0.0013

**Table 2.** Density,  $\rho$ , and density variation with temperature,  $\partial\rho/dT$ , of liquid lead and lithium compared with experimental values.

# Testing Li & Pb potentials V

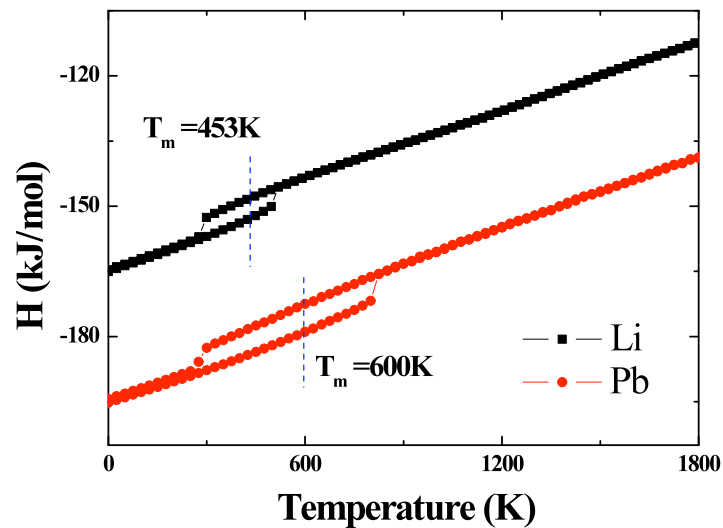
- Heat capacity ( $C_p$ )

$$C_p = \left( \frac{\partial H}{\partial T} \right)_p$$

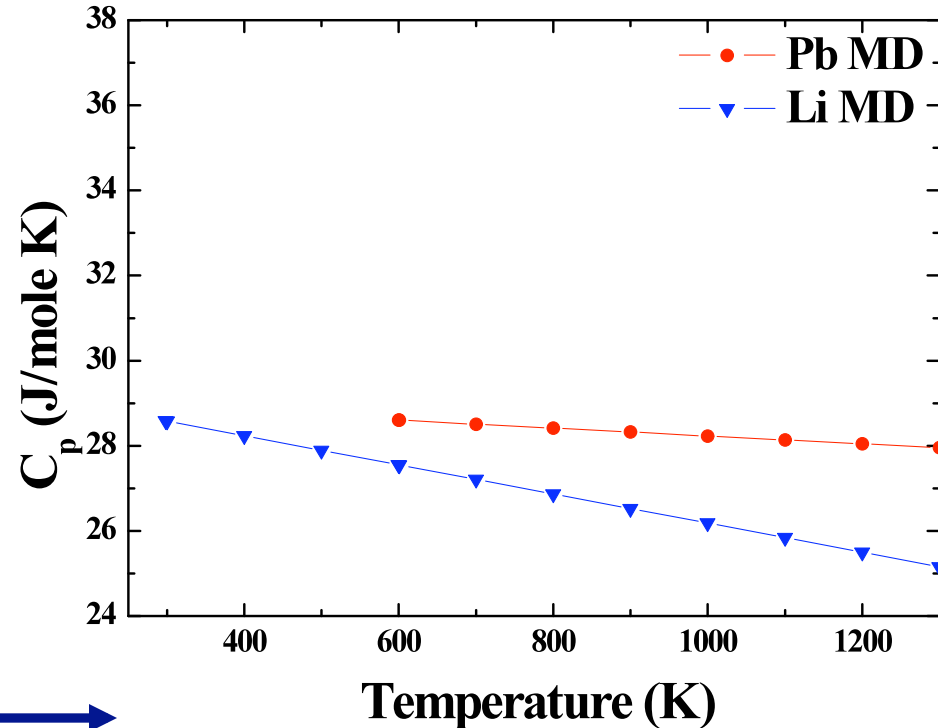


# Testing Li & Pb potentials V

## ■ Heat capacity ( $C_p$ )



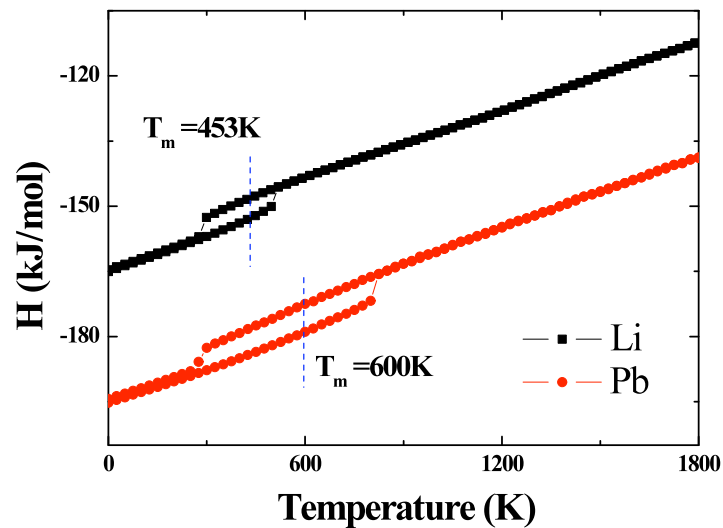
$$C_p = \left( \frac{\partial H}{\partial T} \right)_p$$



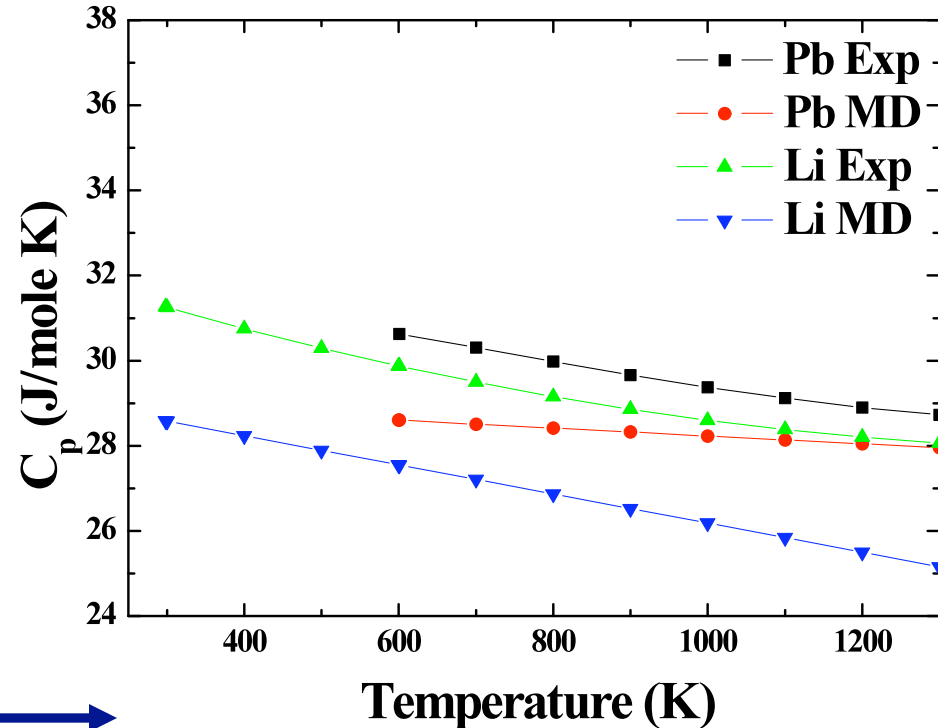


# Testing Li & Pb potentials V

## ■ Heat capacity ( $C_p$ )



$$C_p = \left( \frac{\partial H}{\partial T} \right)_p$$





# Testing Li & Pb potentials VI

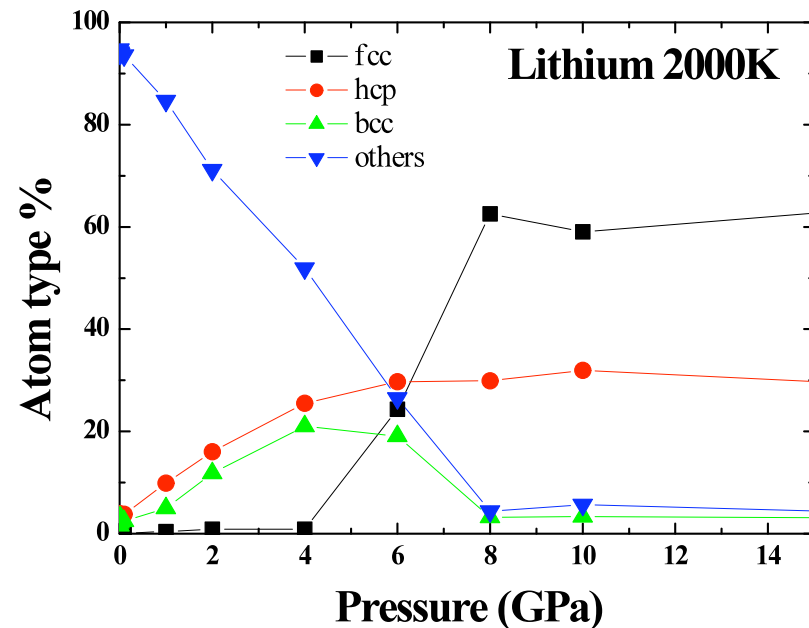
## ■ Pressure

Experimentally, lithium presents a structural phase transition (from bcc to fcc lattice) at high pressures [9] at 7.5 GPa. **Will be our potential good enough to account on that transformation?**

# Testing Li & Pb potentials VI

## ■ Pressure

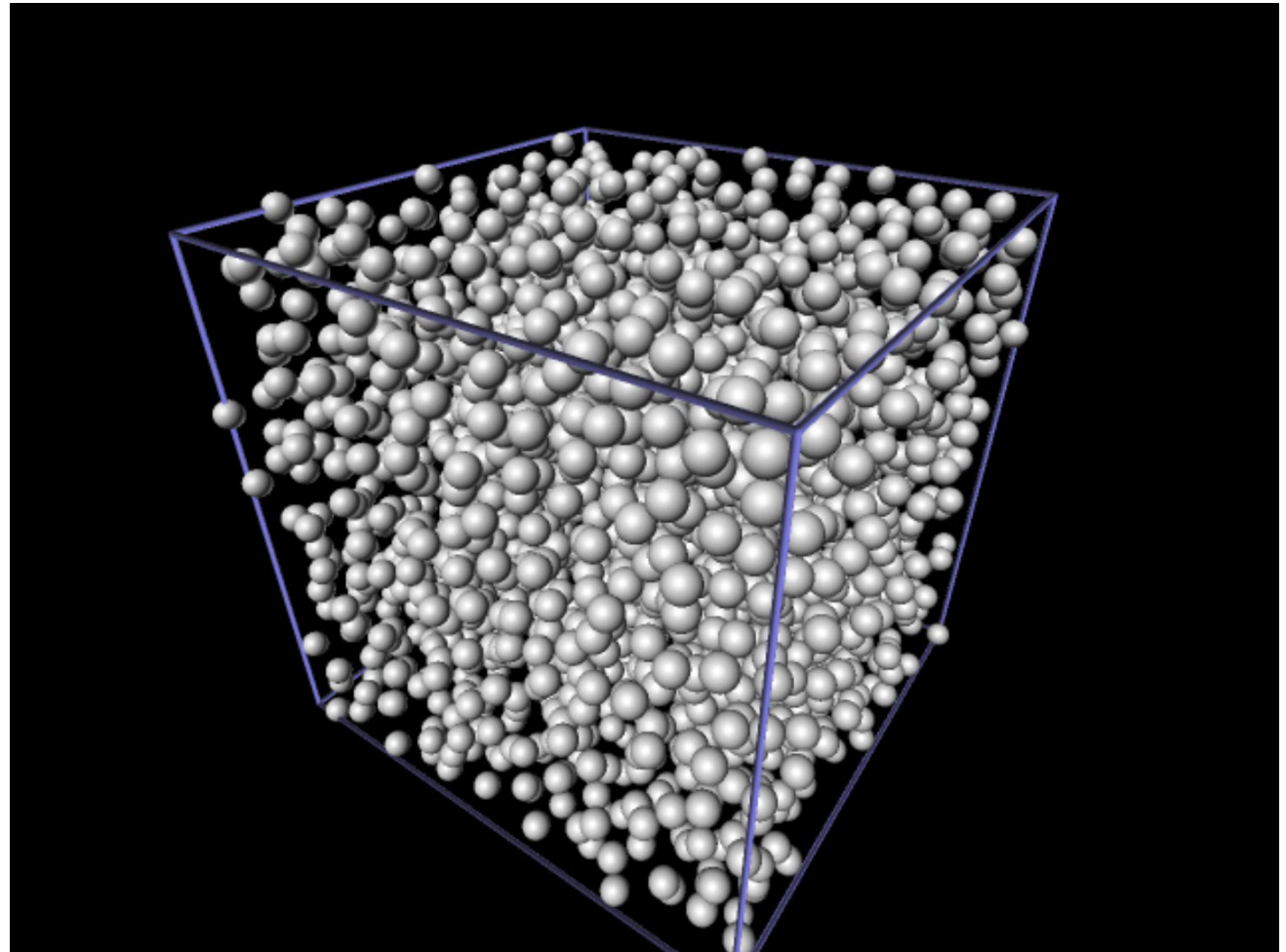
Experimentally, lithium presents a structural phase transition (from bcc to fcc lattice) at high pressures [9] at 7.5 GPa. **Will be our potential good enough to account on that transformation?**



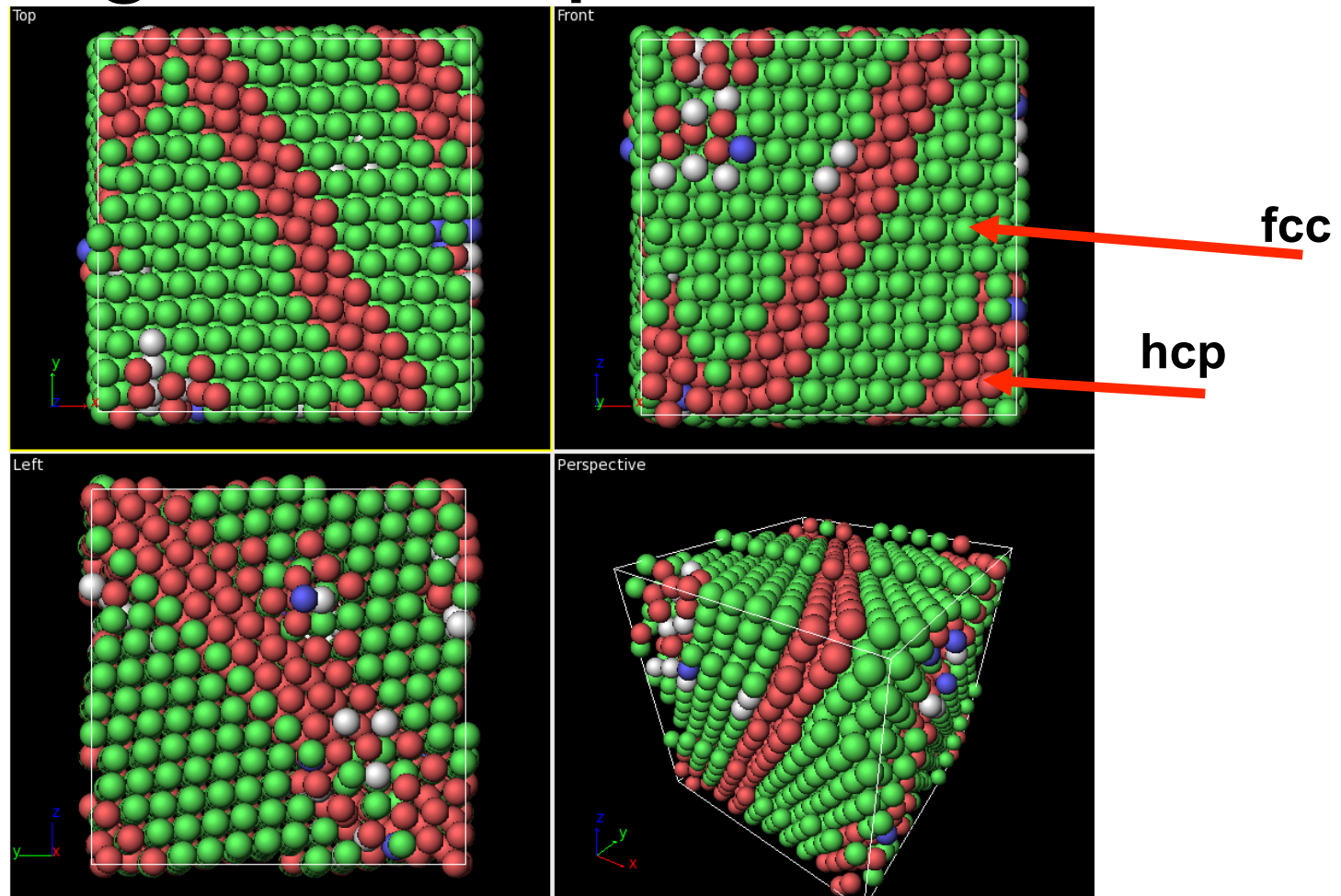
# Testing Li & Pb potentials VI

## ■ Pressure

Lithium  
MD Simulations  
T= 1000K  
Final Pressure  
20 GPa



# Testing Li & Pb potentials VII

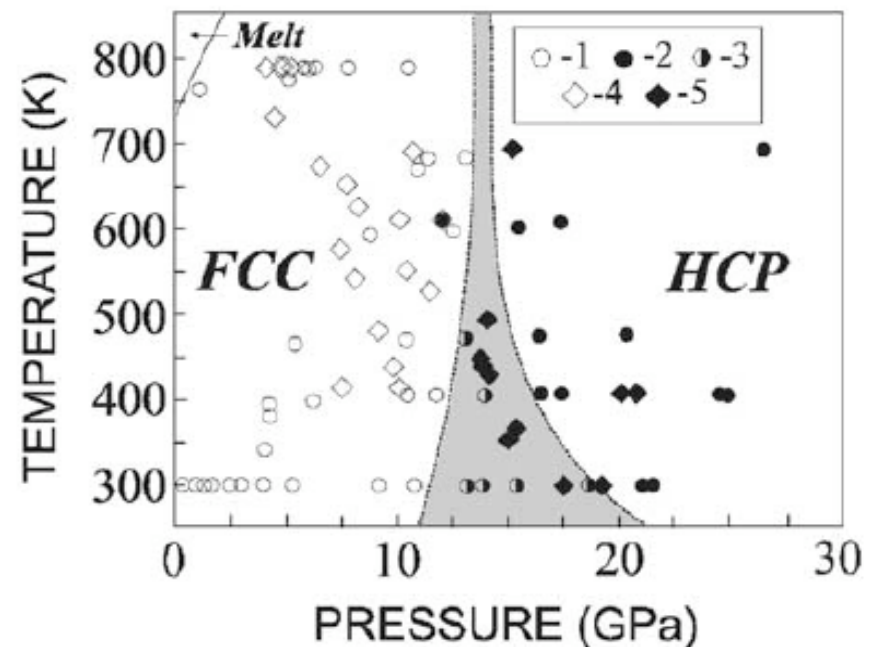


**Figure 4.** Lithium CNA analysis visualization at 2000K and  $P= 8\text{Gpa}$ . ( $N=2000$  Atoms). Green spheres correspond to fcc atoms, red ones to hcp, blue to bcc and white to others.

# Testing Li & Pb potentials VII

Pb crystallizes in the fcc structure under normal conditions. A **transition to the hcp** structure is observed **at 14 GPa** (Takahashi 1969). At about **110 GPa a further transition to the bcc** structure is observed (Mao 1990; Vanderborgh, 1990).

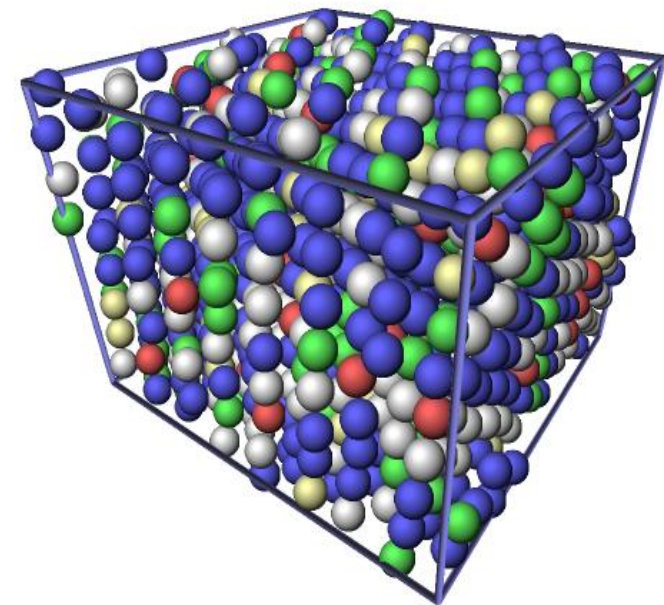
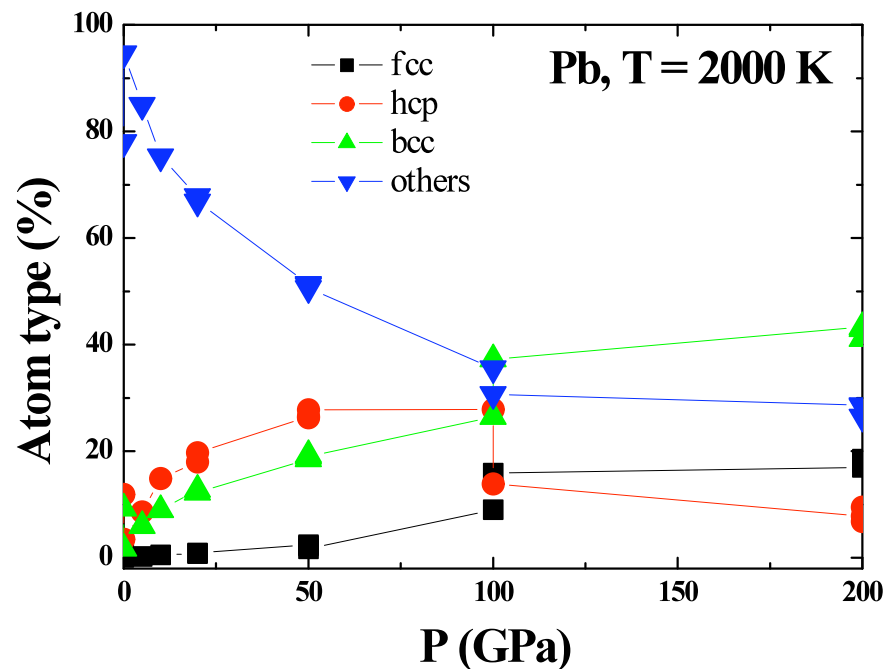
In both cases the volume reduction is very small and there is a large region of **phase coexistence**, consistent with a very small enthalpy difference between the phases over a large pressure interval.



# Testing Li & Pb potentials VII

A transition to the hcp structure is observed at 14 GPa. At about 110 GPa a further transition to the bcc structure is observed.

In both cases the volume reduction is very small and there is a large region of **phase coexistence**, consistent with a very small enthalpy difference between the phases over a large pressure interval [10].



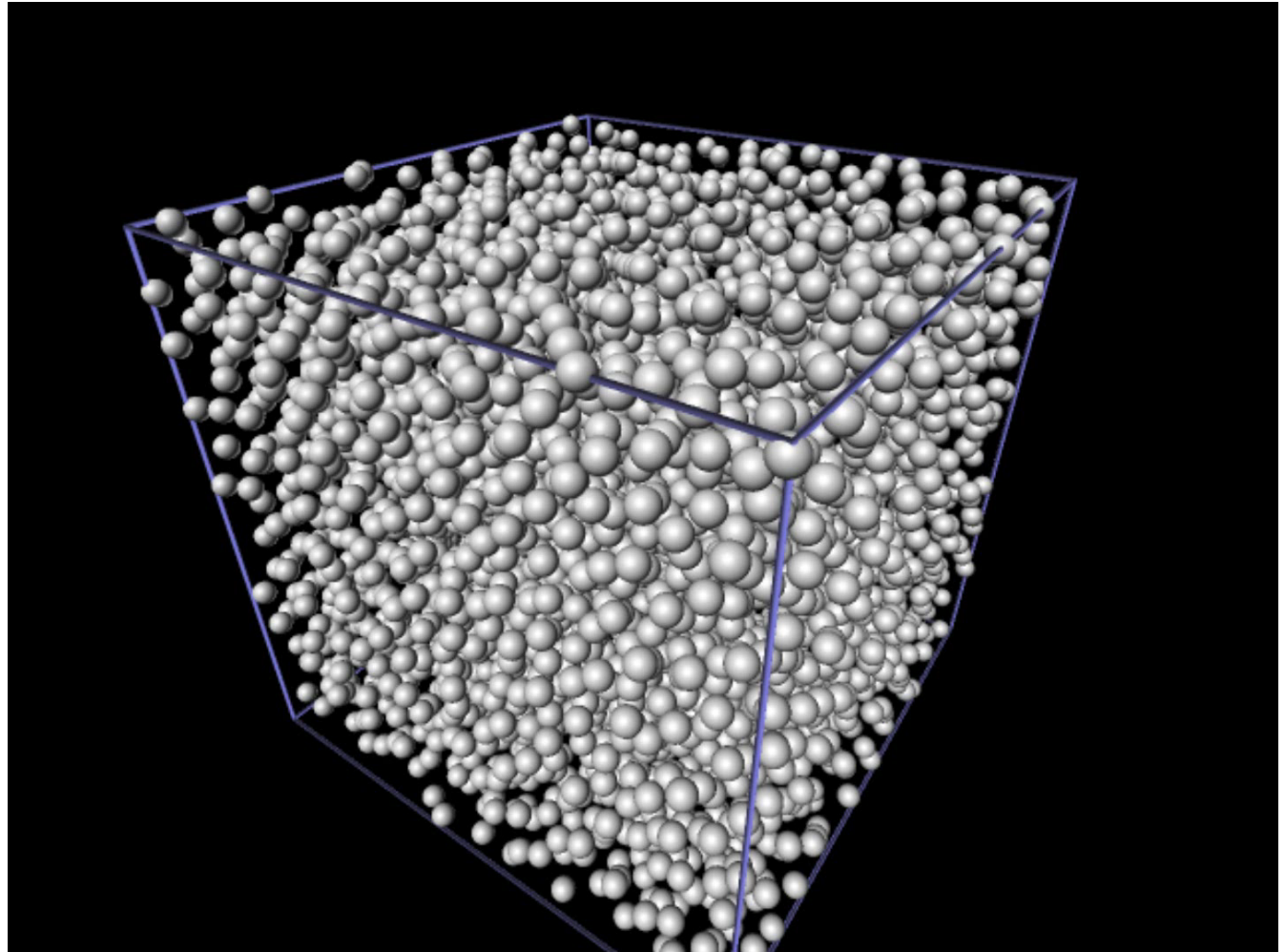
P=200 GPa T=2000 K. Blue = bcc type atoms, green = fcc, red = hcp white = others



# Testing Li & Pb potentials VII

## ■ Pressure

Lead  
MD Simulations  
T= 1000K  
Final Pressure  
200 GPa







- Motivation
- MD, potentials and methods
- Results
- **Conclusions**
- LiPb EAM/alloy/cd results
- Future work



---

# Conclusions

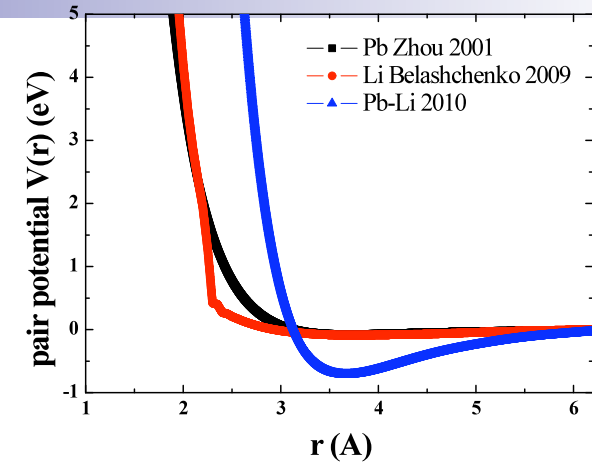
- We trust in our EAM potentials to simulate **liquid Pb and Li in a wide range of temperatures (*and pressures*)**
- ... And to create a new **LiPb cross potential**



- Motivation
- MD, potentials and methods
- Results
- Conclusions
- **LiPb EAM/alloy/cd results**
- Future work

# LiPb cross potential

## ■ EAM/cd



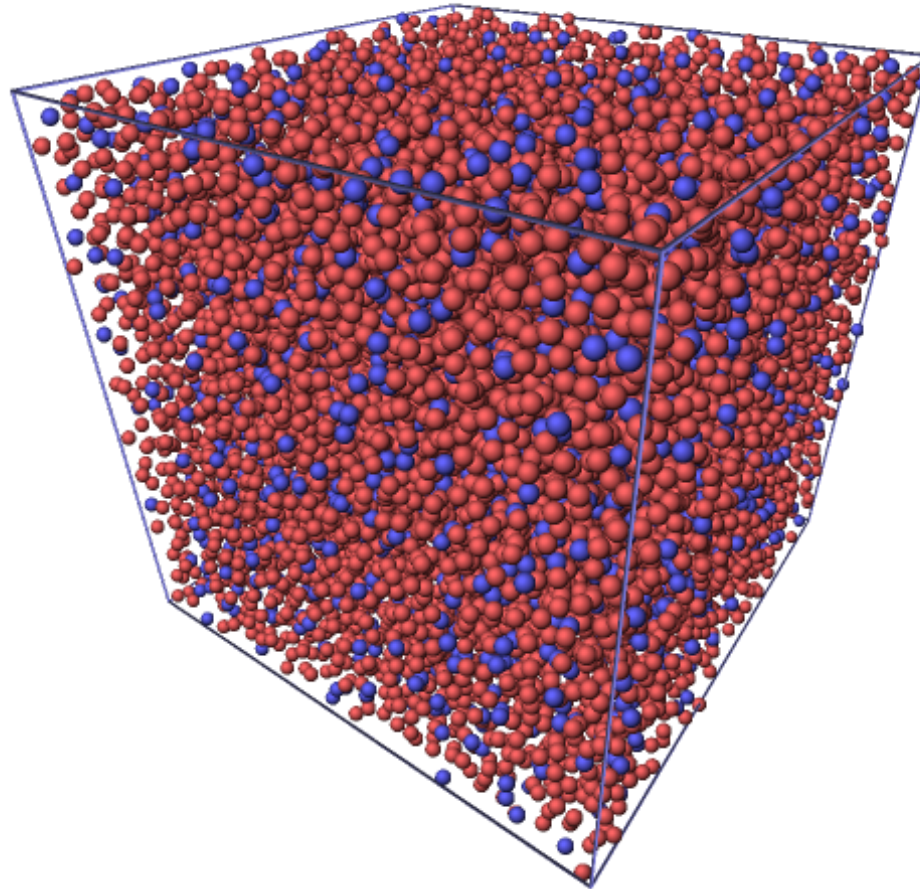
In an EAM model, for a binary alloy system with type a atoms and type-b atoms, there exist two kinds of electron density functions  $\rho_a(\mathbf{r})$  and  $\rho_b(\mathbf{r})$ , two kinds of embedding energy functions  $F_a(\rho)$  and  $F_b(\rho)$ , and three kinds of pair potential functions  $\phi_{aa}(\mathbf{r})$ ,  $\phi_{bb}(\mathbf{r})$ , and  $\phi_{ab}(\mathbf{r})$ .

Usually the six functions  $\rho_a(r)$ ,  $\rho_b(r)$ ,  $F_a(r)$ ,  $F_b(r)$ ,  $\phi_{aa}(r)$ , and  $\phi_{bb}(r)$  are assumed to be **transferable** from monatomic systems to alloy systems.

In our case we have made use of the **effective representation** [11] and develop a LiPb cross potential in the **EAM alloy composition dependent** framework [11].

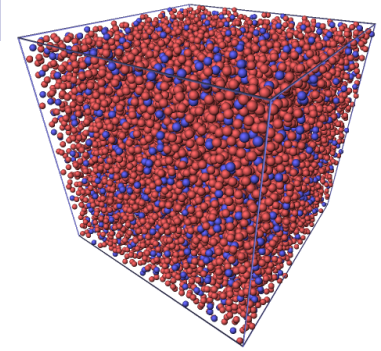
[11] A. Caro, D. A. Crowson, and M. Caro Classical Many-Body Potential for Concentrated Alloys and the Inversion of Order in Iron-Chromium Alloys. PRL 95, 075702 (2005)

# LiPb preliminary results I

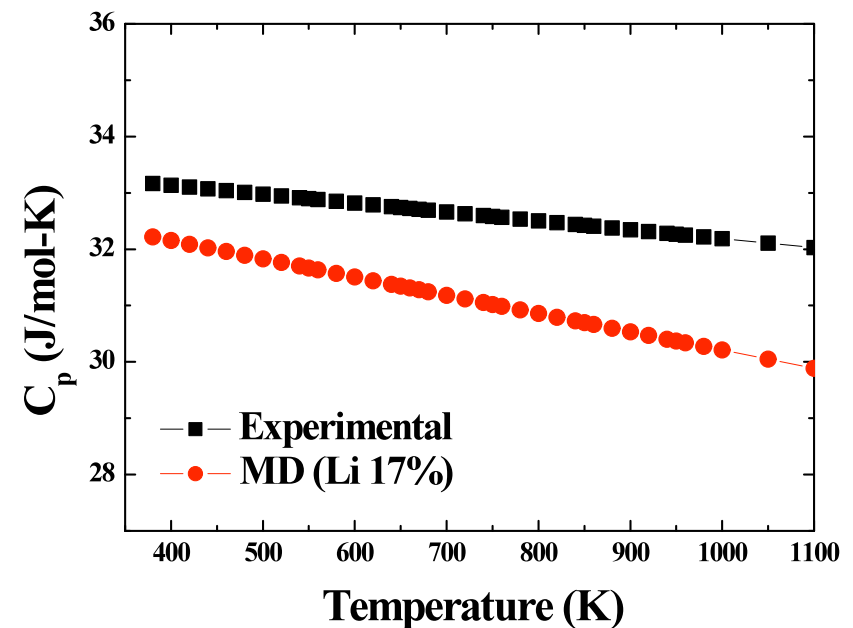
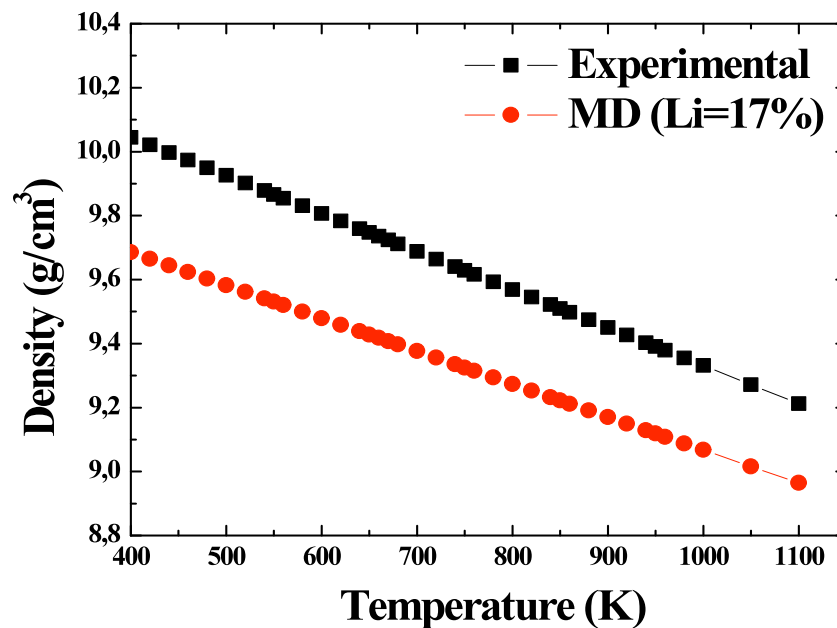


Eutectic LiPb (17%),  $T=1000\text{K}$ ,  $N= 100000$  atoms.

# LiPb preliminary results I



## ■ Eutectic $\rho(T)$ and $C_p$

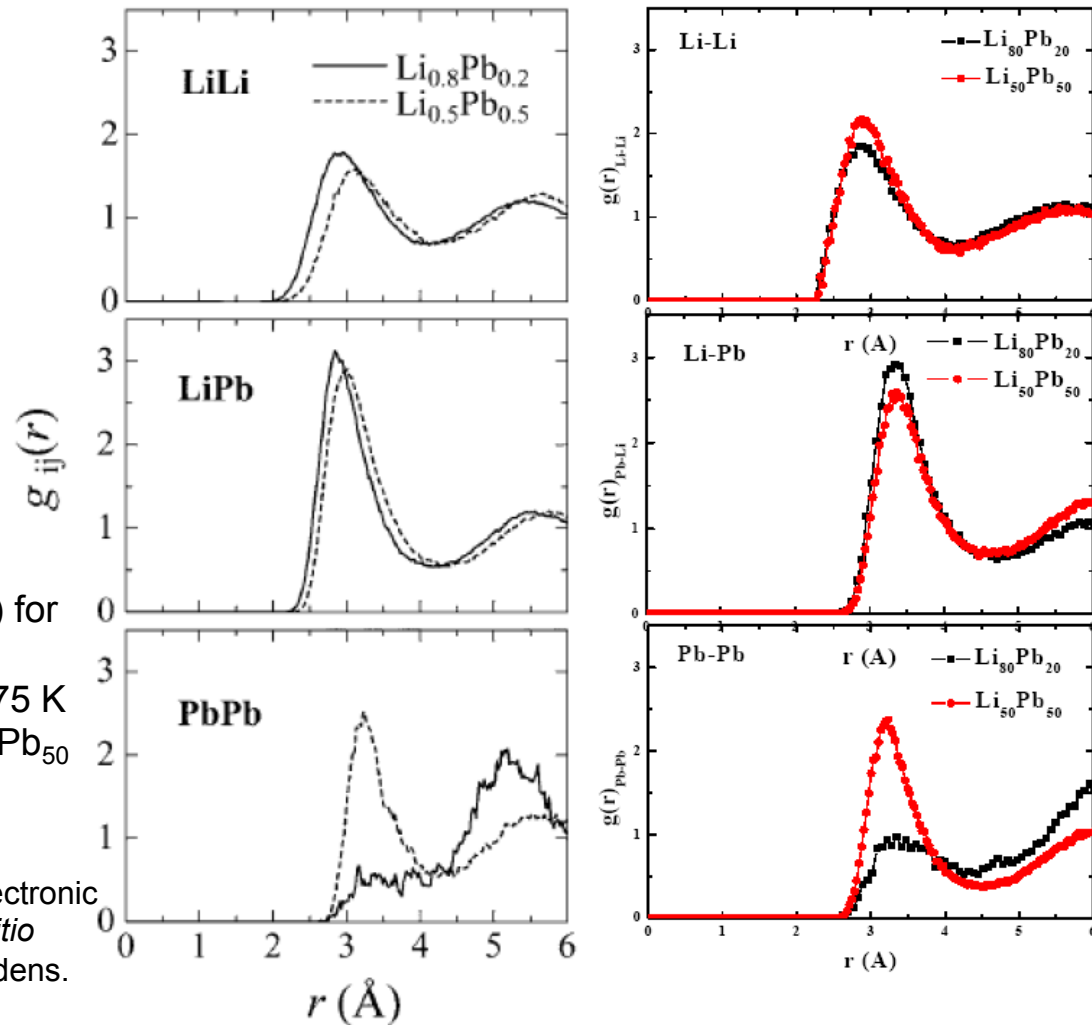


# LiPb preliminary results II

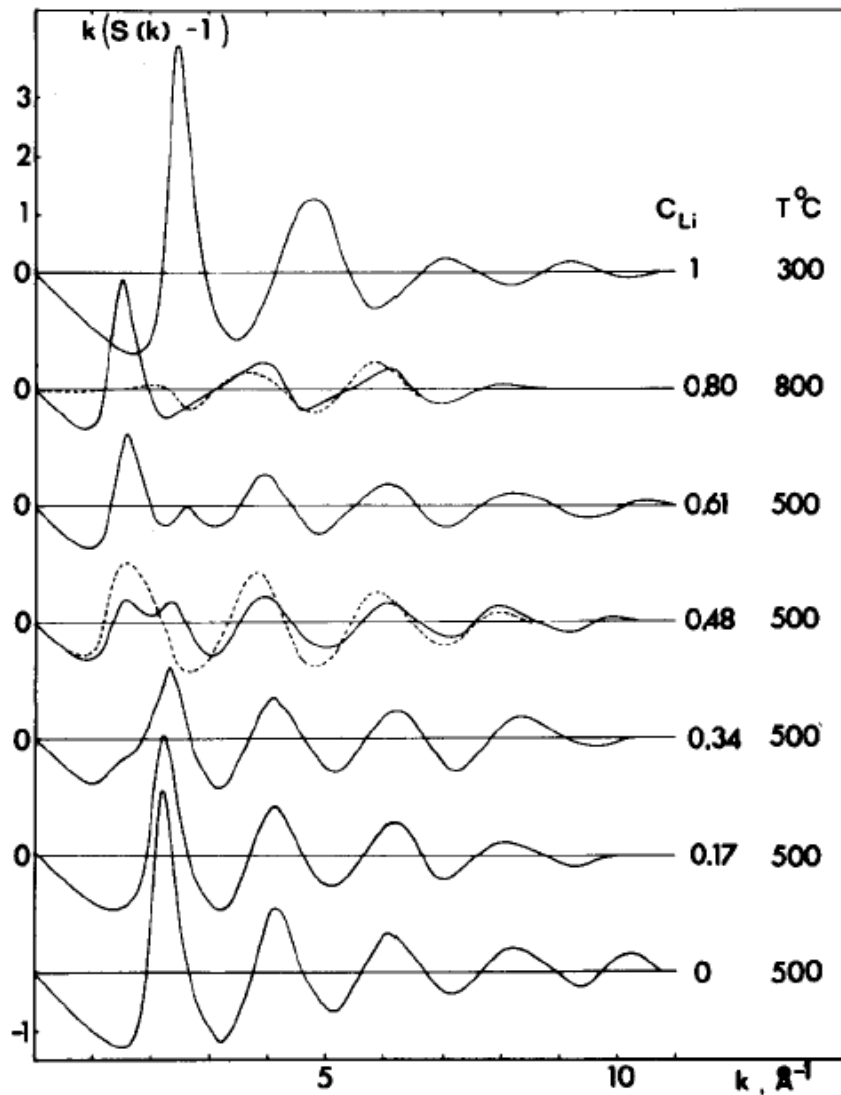
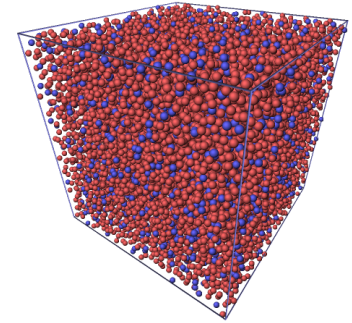
CMD  
VS  
*ab initio* [13]

Figure 5. RDF  $g_{\text{LiLi}}(r)$ ,  $g_{\text{LiPb}}(r)$  and  $g_{\text{PbPb}}(r)$  for the liquid  $\text{Li}_{80}\text{Pb}_{20}$  (solid line - black) and  $\text{Li}_{50}\text{Pb}_{50}$  (broken line - red) alloys.  $T = 1075$  K and  $805$  K for the liquid  $\text{Li}_{80}\text{Pb}_{20}$  and  $\text{Li}_{50}\text{Pb}_{50}$  alloys, respectively [13].

[13] Senda et al. The ionic structure and the electronic states of liquid Li-Pb alloys obtained from *ab initio* molecular dynamics simulations. J. Phys.: Condens. Matter 12, 6101 (2000)



# Structural properties



Li17 T = 800K

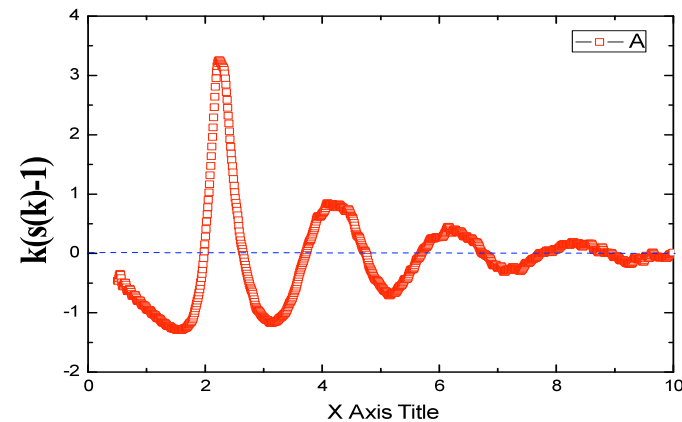


FIG. 4.  $k(S(k) - 1)$  curves of liquid  ${}^7\text{LiPb}$  alloys, obtained by neutron diffraction. -----: 80 Li, theoretical curve for a random hard sphere mixture (rhs).<sup>2</sup> -----: 48 Li, "zero-alloy curve" calculated by separating a rhs  $S_{NN}(k)$  contribution from the measured curve.

[14] H. Ruppertsberg and H. Egger. The Journal of Chemical Physics, Vol. 63, No.1 0, 15 (1975)



# Structural properties

**S(Q) Li<sub>17</sub>Pb<sub>83</sub> OK**

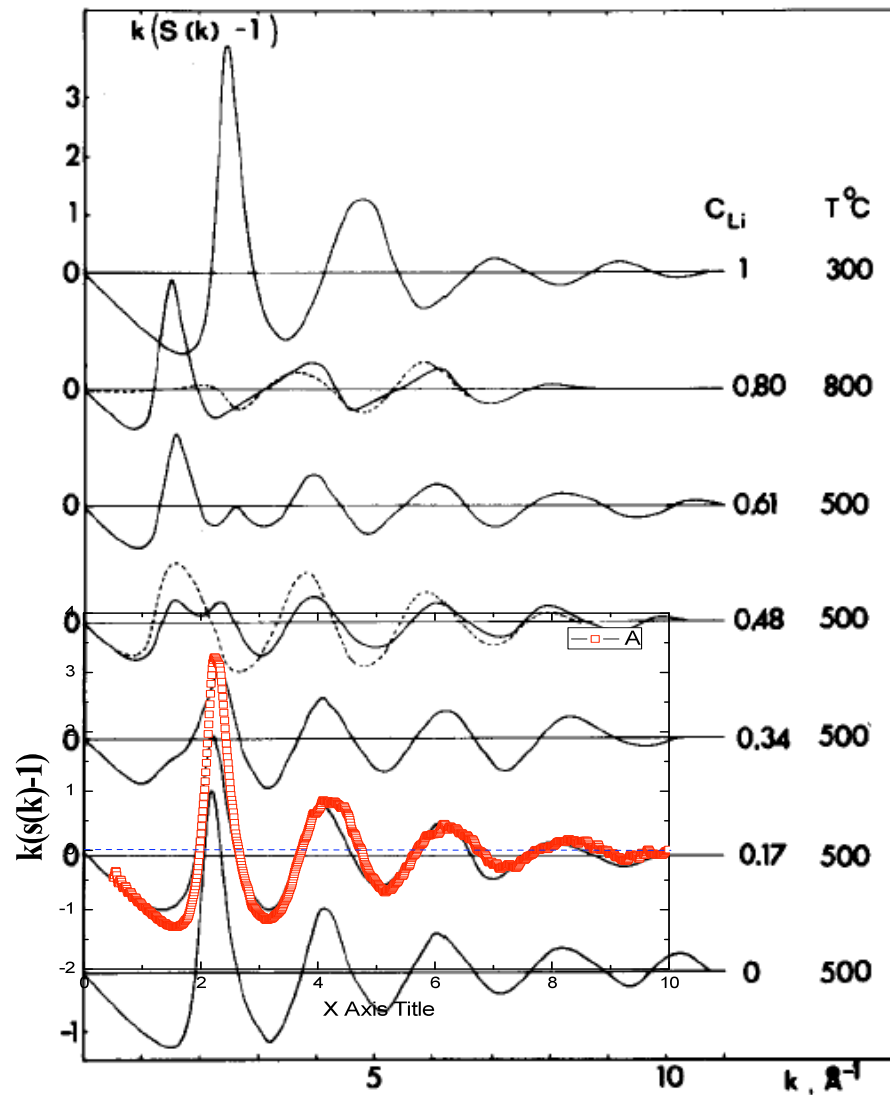
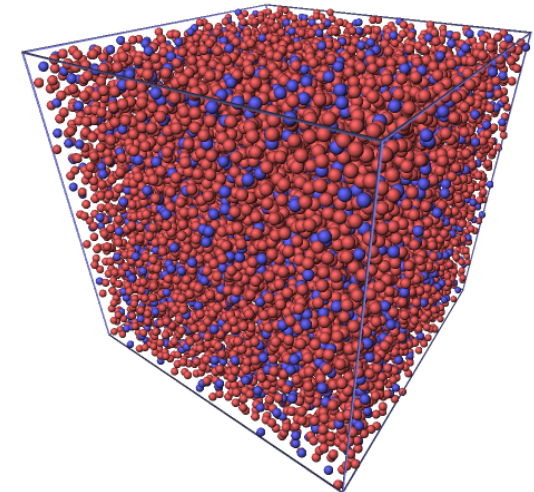


FIG. 4.  $k(S(k) - 1)$  curves of liquid  ${}^7\text{LiPb}$  alloys, obtained by neutron diffraction. -----: 80 Li, theoretical curve for a random hard sphere mixture (rhs).<sup>2</sup> -----: 48 Li, "zero-alloy curve" calculated by separating a rhs  $S_{NN}(k)$  contribution from the measured curve.

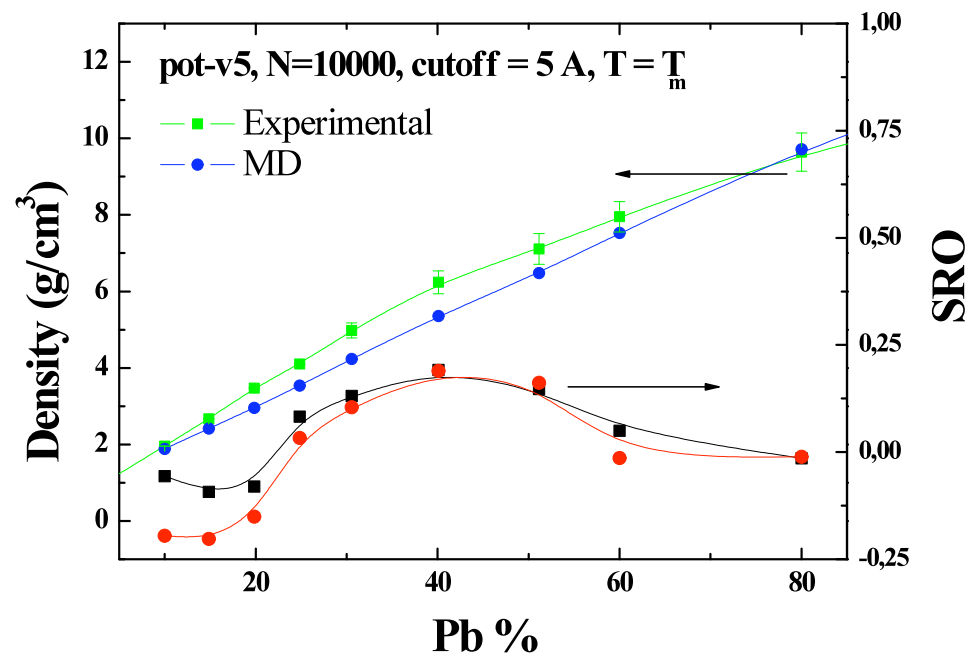




- Motivation
- MD, potentials and methods
- Results
- Conclusions
- LiPb EAM/alloy/cd results
- **Future work**

# Future plans (LiPb)

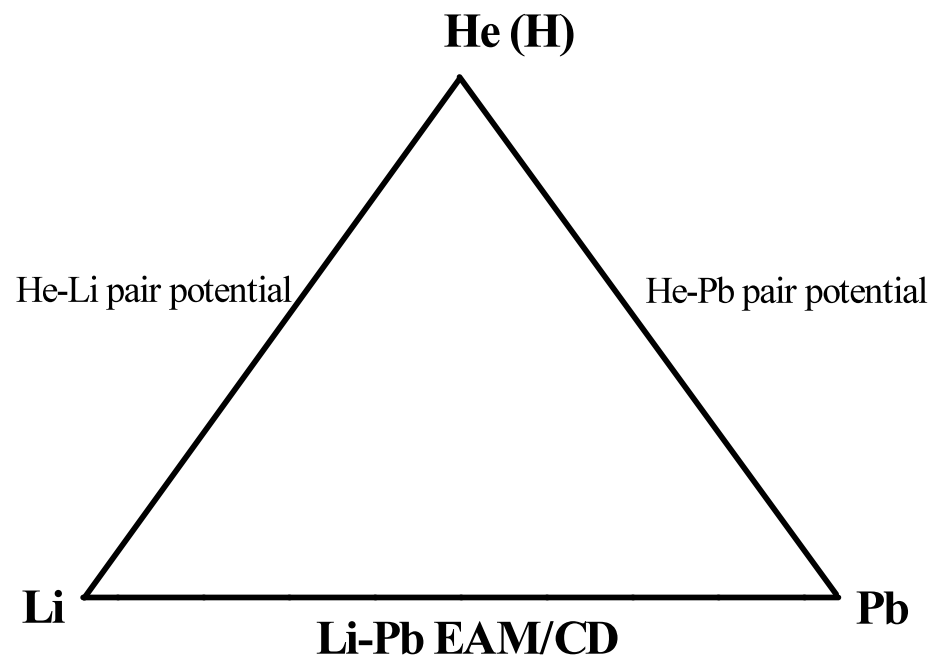
- Final tuning of LiPb cross potential.
- Determine eutectic point.
- Study SRO in liquid LiPb system.



SRO calculated from our MD data. Around eutectic composition SRO is 0.

# Future plans (Li-Pb-He)

- He (and/or T) will be introduced in LiPb.
- He(T)-Pb and He(T)-Li potentials must be developed.



Schematic picture of the ternary potential for Li-Pb-He.



# ACKNOWLEDGEMENTS

- This work is partially funded by the European CONSOLIDER Program.
- The work of the first author is part of their PhD Thesis and has been supported by the *Universidad Politécnica de Madrid* (Spain).
- We are grateful to Professor D. Belaschenko for providing us with his lithium potential.

Thank you for your  
attention

